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Selective Oxide Deposition on Fe-Based Alloys in Dry High Temperature Environment – A Review



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ARTICLE INFO	ABSTRACT
Article history: Received 30 November 2019 Received in revised form 9 March 2020 Accepted 12 March 2020 Available online 30 June 2020	This paper presents a review of an investigation on selective deposition of oxides on Fe-based alloys in dry high temperature environment. The oxidation reaction occurred in order to promote a protective oxide layer to prevent corrosion attack and secure the metal substrate from degrade further. As temperature increased, the oxidation rate was significantly increased. Selective oxidation of an alloy component significantly lowers the concentration of that particular metal element in the alloy subsurface zones. As the cyclic thermal conditions increases, associated mechanical damage to the oxide scale leads to the porous oxide growth and also accelerates the depletion of alloy element. Eventually, prolong the process a point is reached where diffusion of anion to the alloy becomes competitive with the outward diffusion of metal cation to the developed protective scale. Selective oxidation. It is considered a short-path circuit for diffusion and hence leads to weaken the cohesive strength of individual grains. The focus of this work is to review the complex oxidation process due to the occurrence of selective oxidation and oxide deposition with the growth of oxide scale at different rates and also due to the presence of defects and clear microstructure differences of Fe – based alloys. It is also to enhance basic knowledge regarding the oxide layer behaviour that requires an understanding of alloy composition and microstructure. The consequences of oxide growth are specifically relative to the effects of factors such as exposure duration, temperature and operating parameters that must be clearly understood.
<i>Keywords:</i> Fe-based alloys; oxidation; high temperature	Copyright © 2020 PENERBIT AKADEMIA BARU - All rights reserved

1. Introduction

In high temperature service, oxidation becomes the most and prominent corrosion reaction in many industrial environments. In fact, metal alloys often rely on oxidation reaction in order to promote a protective oxide layer to prevent corrosion attack and secure the metal substrate from degrade further [1-3]. Common corrosion attack in high temperature service such as sulfidation,

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carburization and ash or salt deposited corrosion. Generally, when Fe-based alloys exposed to high temperature corrosion or oxidation, the oxide scale formed in the temperature range of $370 - 560^{\circ}$ C was mainly Fe₂O₃. As the temperature increased, the oxidation rate was significantly increased [4]. Particularly, the flow regimes of multiphase fluids or gases extremely have vital effects on the rate of corrosion. In high flow rates regimes, flow-induced corrosion and erosion-corrosion may occur.

Moreover, when the flow rates are low, typical type of pitting corrosion is more likely to occur. It is also related to the velocity amount and nature of sediment and definitely contaminants and dirt from gas applied [5-8]. Instead of that, the combination of gases can greatly accelerate the metal corrosion. In addition to that, the metal surface structure or joint design may become interest part providing sites for corrosion to occur. Importantly, these factors should be taken into account in order to inhibit the internal corrosion degradation and provide adequate method in preventing any accidents and unplanned shutdown [9-12]. Typically, the unplanned shutdown or plant breakdown appeared when involved catastrophic mechanical failures. Initially, when Fe - based alloys especially in piping or structure exposed to high temperature oxidation it provides a clearer and more complex implication [13]. This is because each alloying element in metal composition has different affinity to oxygen, which promotes the formation of different free-energy metals, producing more complex oxide compounds, solid solubility rates between different oxides, the presence of metal ions have different mobility and diffusion, and also include oxygen dissipation which results in sub-surface precipitation involving internal oxidation [14-15]. Based on these factors, the problem associated with oxidation for Fe – based alloys should be given an emphasis in this study. Therefore, high temperature corrosion for Fe – based alloys structure or piping in contact with oxidation become a dominant technology problem and to overcome it, is a challenge that needs to be addressed perfectly.

In this study, the focus is aimed to review the oxidation process which is more complex due to the occurrence of selective oxidation with the growth of oxide scale at different rates and also due to the presence of defects and clear microstructure differences of Fe – based alloys [16-20]. More complex conditions occur when it involves the oxidation of a metal in which the metal itself has different phase changes, composition, microstructure, and presence of defects that contribute to the reduction of performance. Factors such as design, the difference in chemicals and heat treatment plays a critical role in which it can affect the quality of the metal.

2. High Temperature Oxidation in Fe-based Alloys

Oxidation involving Fe – based alloys structures is a common problem faced by various industries. This is because the metal has undergone solidification and transformation in solid state, having an uneven surface, composition, structure and geometry that differ from base metal parts. Oxidation of the metal substrate produces a thin layer of oxide layers that encloses the entire surface of the substrate [21]. This involves a series of oxidizing mechanisms such as transporting or migrating oxygen molecules by physical adsorption to metal surfaces, chemical absorption, nucleation and oxide growth. In several types of chemical bonding usually it involves electron donation and acceptance especially during oxidation. Mostly metal oxides, sulphides etc. shows to have ionic bonding and has definite crystallographic structure where distributions of anions and cations was found at different distinct locations. As well discussed earlier, oxides usually consist of grains like metals, in crystallize condition, appear in grain growth. Basically, when it exposed to high temperature, the oxide itself may plastically damage and deformed and at high temperature it may plastically deformed [22].



Commonly, metal oxides present good conductivity properties and falls under semiconductors class category. It means that their conductivity falls between the insulators and metallic conductors [23]. This type of conductivity property increases with a slight shift from stoichiometric proportions of metal and oxygen. Generally, it happens when the temperature increased. Additionally, there are two types of semiconducting oxides which are; (i) p-type (positive carrier) and n-type (negative carrier). The p-type oxides include Cu₂O, NiO, FeO, Cr₂O₃ and Fe₃O₄. This type of oxides shift of stoichiometric proportions takes the form of a certain number of missing metal ions in the oxide lattice called cation vacancies [24-27]. It also maintains the electrical neutrality. The other type is n-type oxides which includes ZnO, CdO, TiO₂, Al₂O₃ and Fe₂O₃. For this type, excess of metal ions exist in interstitials positions of the oxide lattice. In addition to that, these metal ions suggested migrating with the electrons to the outer oxide surface during the oxidation takes place [28].

3. Selective Oxide Layer Compound

In many commercial production industries, the oxidation in environment containing free oxygen, carbon dioxide gases, water vapour and nitrogen takes place. Furthermore, the oxidation mechanism in ambient is definitely differs from that in a mixed-gas atmospheres [29-32]. On the other hand, when it involves metal alloys, it usually contains impurity elements and other added alloying elements. Each of the alloying elements presents different behaviour from each other. Typically, the presence of Cr, Al and Si which definitely less noble than Fe, contribute a certain level of oxidation resistance [33]. This would lead to insignificant protection effect because their levels are very low. Yanagihara *et al.,* [34] reported peaks detection of Fe, FeSiO₄ and FeO was found on Fe-Si alloy annealed in atmosphere with a dew point of 71°C and having an oxygen potential higher than the equilibrium oxygen potential of the Fe-FeO system. It is demonstrated that the oxidation of Fe-Si alloy agrees very well with the prediction and confirmed the Si concentration was high according to the existence of FeSiO₄. Presumably, the FeSiO₄ phase detected was mainly in the internal oxidation zone [35].

Generally, when Fe oxidized, the familiar result observed is rust. The main compounds formed during reaction between Fe and oxygen when in contact with plenty of oxygen around. The compound form in Eq. (1) is called iron (II) oxide or ferrous oxide. This type of oxide is categorized under inorganic compound with the formula FeO. It is also known as wustite and presents in black-colored powder [36]. Then, if Fe and oxygen are in contact at high enough temperatures, the iron (III) oxide, Fe₂O₃ formed and the common name is hematite. Hematite presents in grey or reddish compound as shown in Eq. (2). When Fe is heated in the presence of less oxygen, it is possible to synthesize to form another iron oxide, Fe₃O₄ called magnetite as shown in Eq. (3). The magnetite occurs in black color.

$$2Fe(s) + O_2(g) \to 2FeO(s) - wustite \tag{1}$$

$$4Fe(s) + 3O_2(g) \rightarrow Fe_2O_3(s) - hematite$$
⁽²⁾

$$3Fe(s) + 2O_2(g) \rightarrow Fe_3O_4(s) - magnetite$$
 (3)

Basically, Iron oxides are possibly reduced by graphite during high temperature exposure. They revealed that the reduction of mass proved to present as an effect of the reduction of Fe_3O_4 and Fe_2O_3 to Fe and FeO [37]. The distinct reaction was presumably to be indirect reduction. It is mainly resulting from the reduction of nano-sized graphite with water that comprised in the argon gas.



Larsson *et al.*, [38] found after 1-hour exposure in dry oxygen at 600°C, a thin layer of hematite deposition was dominated and covered, then become a thicker formation of magnetite layer on the surface of pure iron. Meanwhile, wustite formed at the interface between metal and oxide region. Besides, the oxide properties found on the pure iron revealed that after exposure to 1 and 3 hours, the porosity occurs and was found closely to the hematite interface. As for comparison, after 24 hours distribution the porosity was found more centered especially in the layer of magnetite. Marcius *et al.*, [39] used Raman spectroscopy to show the presence of hematite in the outer oxide layer on iron plate exposed at 500 and 600°C. They revealed that at low temperature of 300 and 400°C composed dominantly with magnetite and in a small fraction of hematite. Hisa *et al.*, [32], Larsson *et al.*, [38] and Marcius *et al.*, [39] studied and revealed on the temperature and time dependence leads to different morphologies of iron oxide products. In addition to that, Marcius *et al.*, [39] explained deeply on the hematite morphologies. They found that hematite particles present in elongated and very thin suggesting the shape of corn plant leaves. As temperature increased, the particle size increased perpendicular to their elongation [39-40].

As reported by Bertrand *et al.*, [12], Grosvenor *et al.*, [29], Larsson *et al.*, [38] and Marcius *et al.*, [39], the scale structure developed on Fe-alloys changes readily depending on the atmosphere and temperature, and most likely the result from structural change and scale formation was definitely found to affect the phase transformation behaviour of FeO. In addition to that, Kondo and Tanei [41] proved on the FeO phase transformation by oxidized the ultra-low carbon steel sheets in the furnace and held at 550°C or 700°C for 30 min to have a scale of different structures before phase transformation. He found that the bilayer scale of Fe₃O₄ and FeO transforms from the Fe₃O₄/FeO interface, while the monolayer scale of FeO alone generates Fe₃O₄ precipitates at the scale/steel interface preferentially. He also suggested that the phase transformation behaviour of FeO can be controlled by the initial scale structure. According to that, Kondo and Tanei [41] was found in agreement with Bertrand *et al.*, [12], Grosvenor *et al.*, [29], Larsson *et al.*, [38] and Marcius *et al.*, [39].

In the context of designing a corrosion resistance for components or structures, the fundamental aspect is basically depends on the formation of a slow-growing and protective oxide scale. It is developed mainly by selective oxidation of an appropriate alloy component [42,38]. Generally, selective oxidation process of an alloy component significantly lowers the concentration of that particular metal element in the alloy subsurface zones. As the cyclic thermal conditions increases, associated mechanical damage to the oxide scale leads to the porous oxide growth and also accelerates the depletion of alloy element [43-45]. Eventually, prolong the process; a point is reached where diffusion of anion to the alloy becomes competitive with the outward diffusion of metal cation to the developed protective scale. In high temperature environment with oxygen-rich atmospheres, resultant iron oxide scale which basically presented on the metal surface. As the oxidation prolong, it is sufficient to induce internal oxidation and it may proceed inside the grains and along the grain boundaries [46-49]. Selective oxidation changes the mechanical properties mainly due to the grain boundary oxidation. It is considered a short-path circuit for diffusion and hence leads to weaken the cohesive strength of individual grains. It is also influenced mainly by the temperature gradient that caused severe embrittlement.

In addition to that, Yuan *et al.*, [50] discussed on the effect of Si content on different low carbon steels at heating temperature of 1260° C. They found that Si appears to distribute in two distinct forms which are, (i) Fe₂SiO₄ (fayalite) where it takes place in the innermost region of the oxide scale, and (ii) granular SiO₂ that was dispersively distributed mainly in the matrix closely near to the scale region. They reported that for steels with low Si content, less penetration of Fe₂SiO₄ specifically found along the grain boundary. This happens because it contains small amount of Fe₂SiO₄ and leads to the



unnoticeable net-like pattern of Fe₂SiO₄. Meanwhile, for low carbon steel with 1.21 wt.% Si, the netlike pattern of Fe₂SiO₄ can be simply detected mainly in the innermost region of oxide scale which closely next to the Fe-matrix [50,28].

4. Oxide Layer Properties

Alloy metal for high temperature applications depends on the construction and development of protective oxide layer formation to prevent high temperature effects. A good protective oxide layer has properties such as growing and formed at a steady rate and having a strong adhesion. Most of the metal alloys used in high temperature applications maintain resistance to high temperature corrosion by producing a protective oxide layer composed of alumina layers (Al₂O₃), chromia (Cr₂O₃) and silica (SiO₂) or spinel [51-54].

The oxide growth at temperature between 150 and 350°C could be described to have a homogenous and planar FeO layer. At this temperature Fe cations act as moving species during oxide growth and the oxidation rate is limited. Thin oxide films enough to cause interference colours until the film thickness reaches about 3 x 10-6cm and at the initial stage, it is invisible to ordinary observation [55]. Abuluwefa [1] performed an experiment for carbon steels samples and it shows that oxidation that occurs at lower temperature presents approximately a linear type of oxidation especially at temperature of 1000°C. It shows the result of the controlling mechanism of oxidation. Yet, since it was a very thin oxide layer without any resistance barrier for oxygen/iron ion transportation to the reaction surface, hence it is exhibited a linear behavior [56,1].

Meanwhile, it differs when compared to porous oxide scale growth. Prolong the time of exposure to certain temperature, oxide layer undergone thickening process. Thicker oxide layers with thickness more than 3000Å are called scales and its porous behaviour does not act to prevent access of oxygen to the metal surface. It usually composed of several layers of different oxides and commonly above 566 °C, Fe-scales consists of three layers which are (i) FeO developed adjacent to steel base metal, (ii) Fe₃O₄ located in the middle layer and (iii) Fe₂O₃ that found on the top of surface layer [57-60]. It has been known that oxide scale on steel surface often detaches from the base metal during high temperature process. The detached and swelling oxide scale condition is called blistering. Blistering is one of the surface oxide scale defects that usually found during hot rolling process and it is occurred because of the growth stress due to scale formation. Hwan Bak *et al.*, [36] reported that unprotective oxide scale developed on Fe-2%Mn-0.5% Si-0.2% welded joint was found to be thicker on the outer scale compared to the inner scale. The rough and thick oxide scale was found susceptible to cracking and detached off from the inner mixed scale due to the large compressive stress developed. They found that the oxide scale was quite thick, brittle, cracked and could detach easily [61].

As reported by Dulgosch *et al.*, [62] the low-carbon steel nanowire samples with thickness of 200 μ m were subjected to an oxidation in ambient air for 0.5 – 2 hours at temperatures of 300°C – 700°C. At initial state at 300°C, no significant changes and oxide layer was clearly homogeneous. Furthermore, it shows an increase to 500°C for 2 hours and significantly enhances the number of hematite nanowires. As it reached 600°C, elongated and dense coverage with dominations of nanowires and nanoflakes is distinctly detected. It is indicated that at 700°C the well-known top hematite layer. Meanwhile, the underneath layer of magnetite is visibly detectable in the form of floral-like arrangements of nanowires. In addition, the top layer of hematite occurs in the presence of porous network. It can be said that the growth of nano-crystalline iron-oxide indicates the influence of oxidation temperature [62-63].

It is not in agreement with Hwan Bak *et al.*, [36] and as reported by Abuluwefa [1] the higher values of carbon content lead to the lower overall oxidation behaviour of steel, especially when



exposed to high temperatures. He suggested that in higher levels of free oxygen atmospheres, development of oxide layers which comprised of three phases, mainly wustite FeO, magnetite Fe3O4 and a very small layer of hematite Fe₂O₃ was clearly dominated as the oxide products. It is suggested that to control scaling rates of the carbon steels, it is required to control the level of free oxygens in the furnace atmosphere. Instead of Abuluwefa [1], Suarez *et al.*, [64] and Dulgosch *et al.*, [62], Trindade *et al.*, [65] also stated the similar findings on the possible phases formed. They stated that at 550°C, in the region of higher oxygen potential the oxides formed Fe-Cr-O are hematite (Fe₂O₃) and chromia (Cr₂O₃). Meanwhile, for lower oxygen potentials, oxide products such as magnetite (Fe₃O₄) and spinel (FeCr₂O₄) are distinctly formed. Oxide product like Wustite (FeO) is not expected to occur since it is not stable below 570°C. They confirmed the scale composed of a coarse-grained outer scale (Fe₂O₃ + Fe₃O₄), and it appears to grow by outward iron diffusion. For development of a fine-grained porous inner scale, it was found that the growth was dominated by inward oxygen diffusion (Fe₃O₄ + FeCr₂O₄) [65-68].

As compared to those observed by Kondo and Tanei [41], Yates et al., [71], Yanagihara et al., [34] and Yu et al., [69] observed more on the microstructure and morphologies of oxide growth on the typical steel substrate as indicated in Figure 1. As shown in Figure 1(a) and (b) Electron backscattered diffraction (EBSD) phase mapping showing a columnar-shape microstructure at the region between the outer granular grains and the globular inner layers. They found that the oxide scale developed consisted of a thin outer layer of hematite (Fe₂O₃) and the inner duplex magnetite (Fe₃O₄) layers [69,71-74]. It is confirmed by EBSD that the outer layer is existed in columnar structure, while the inner layer is much finer grained and the grains appears in equiaxed. Moreover, the grains of magnetite show to have granular shape with the size around 3µm and mostly found in the outer layer of oxide scale. In addition to that, hematite was found closely to the surface and gradually penetrates into the cracks within the oxide scale. Yu et al., [69] indicates some differences compared to the other studies and they detailed out the shape and the sequence of the oxide outer and inner layer developed on the steel substrate. It shows that the diffusion of metal ions which can leads into vacancies and cavity, in order to facilitate the local pores formation [69-75]. They found that the grain-refined metal substrate possibly can increase the grain boundary diffusion specifically at high temperature.

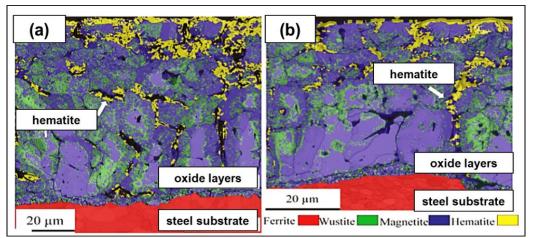


Fig. 1. Phase maps using EBSD of the microalloyed steel at 860°C and then cooling rates of (a) 10%, 10°C/s, and (b) 13%, 23°C/s [69]

Yuan *et al.*, [50] discussed the oxide growth of nanowire by the stress-drive mechanism. Although it is still remains in unclear grey area on how the compressive stress is generally developed in the



oxide layer, but the volume difference between metal and oxides simply shows clear reflection especially in the development of scale thickness. As indicated in Figure 2, since wustite (FeO) does not form below 570°C, therefore Fe oxidized below this temperature and is expected to form a twolayer structure of Fe₃O₄ (magnetite) and Fe₂O₃ (hematite) with Fe₃O₄ next to the Fe substrate [76-80]. When the temperature reached above 570°C, the oxide layer sequence is FeO, Fe₃O₄ and Fe₂O₃ with the FeO layer developed next to Fe. According to the Figure 2, mechanism of gas oxygen diffusion into the oxide lattice during growth of oxide was first explained by chemisorption process on the Fe₂O₃ surface and definitely by attracting an electron from a Fe lattice site, and thus forming an electron hole. Then, the fully ionized chemisorbed oxygen forming another hole and a Fe³⁺ ion enters the surface to find the O²⁻ partner [81-83]. Therefore, another hole and vacancy formed mainly in the cation sub-lattice location. Subsequently, Fe cation vacancies and electron holes are distinctly created at the Fe₂O₃ surface. Following the process, this stage occurs when O₂ at Fe₂O₃ surface utilized Fe³⁺ ions from the outer Fe₂O₃ lattice in order to form new Fe₂O₃ molecules [84].

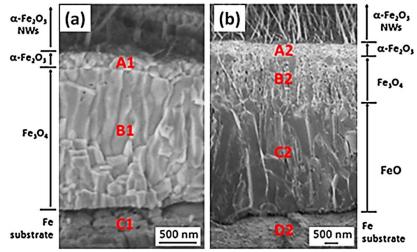


Fig. 2. SEM cross-sectional microstructural images of Fe substrates exposed to oxidation at (a) 400°C and (b) 600°C [70]

As reported by Hwan Bak *et al.*, [36] and Yuan *et al.*, [50], they were in agreement with Yuan *et al.*, [70]. Yuan *et al.*, [70] stated that the Pilling-Bedworth ration (PBR) of Fe oxide or Si oxide is more than 1 and was found at a temperature of 1260°C. As the low carbon steel with different content of Si were heated at 1260°C. The results show that the PBR is greater than 1, and this happens because the volume of oxide is larger than that of the consumed metal and thus leads to a compressive stress in the oxide. This also related with the pressure difference in the liquefied Fe₂SiO₄ phase at 1260°C, which shows a part of Fe₂SiO₄ that permeate into the inner scale. It was predicted that the liquid Fe₂SiO₄ phase distributes along the FeO grain boundary and the net-like Fe₂SiO₄ phase forms after its solidification. They also suggested that a larger compressive stress due to more Fe₂SiO₄ in steels with a higher Si content. This resulted in a deeper penetration layer [70,85-88].

Instead of Yuan *et al.*, [70], Birosca *et al.*, [14] explained that the oxide scale developed on low carbon steel after exposed to high temperature environment in free flowing air is however is not straightforward. As shown in Figure 3(a) wustite presence in dark colour relative to the magnetite and haematite where appeared in the lightest area. They also found that the wustite grain size increases with increasing temperature, causing grain and grain boundaries number to decrease with increasing temperature. In addition, they observed that the cracks and porosity may have a considerable effect on oxide grain shapes [89].



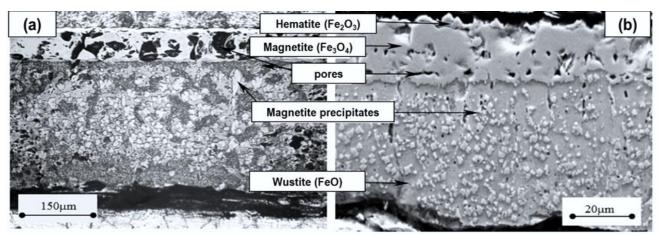


Fig. 3. Oxide scale developed on low carbon steel etched with 1% HCL and ethanol, (a) optical micrograph at 1100°C for 1200 sec, and (b) SEM image at 900°C for 1800 sec [14]

5. Oxidation Mechanism on Iron Alloys

These sections focused on a brief discussion on the oxidation mechanism that related to the previous work which involved Fe-based metal and Fe-alloys. Suggested and proposed mechanism were discussed and explained clearly in order to understand how the oxidation process takes place. As indicated in Figure 4 Suarez *et al.*, [64], they show the scale growth mechanism under controlled and uncontrolled condition. It shows that at short oxidation times, the scale layer is not uniform and adherent. In all cases it is confirmed that wustite is the major oxide layer constituent.

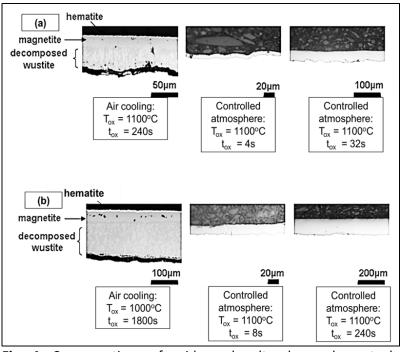


Fig. 4. Cross sections of oxide scale ultra low-carbon steels examined by OM in controlled and uncontrolled condition [64]

Yanagihara *et al.,* [34] illustrates the change of surface structure of Fe-Si alloy due to oxidation. Internal oxidation takes place depends on the solute element due to its outward diffusion and oxygen due to inward diffusion. In the situation where oxygen potential is low, external oxidation becomes predominant. They proposed an illustration of Fe-Si alloy surface covered by SiO₂ films by more than



a certain proportion at the initial stage of oxidation [90-94]. In addition, internal oxidation in such portion is supressed in later stage. Hence, external oxidation only will advance locally while internal oxidation will take place in other parts as shown in Figure 5(a).

In comparison as shown in Figure 5(b), when the oxygen potential of the atmosphere is high, Si forms fine external and internal oxides at an early stage of oxidation and SiO₂ does not form on the surface. Eventually, internal oxidation will advance thereafter homogenously in the depth direction. It can be said that, the higher the oxygen potential of the atmosphere, the larger the amount of internal oxides becomes. As shown in Figure 6 internal oxides content was larger in sample B compared to sample A. Instead of that, surface of sample B was smoother than sample A. As proposed by Yates *et al.*, [71] it is suggested that alloying element and temperature dependance in Kovar plays an important role in determining the oxide layer properties as shown in Figure 6. The layered oxide structure appears to compose of (Co, Fe, Ni)₃O₄ with varying ratios and Fe₂O₃. They confirmed using Raman spectroscopy and XRD that the first top layer initially to consist of CoFe₂O₄. The second layer shows to have the lowest Fe:O ratio and it is suggested to compose of Fe₂O₃. It is thought the third layer to be a (Fe, Co)₃O₄ as confirmed by SEM, STEM and EDX with varying Co content but without Ni [95-96]. The fourth layer was found similar tot third layer but with addition of around 3 at. % Ni and it is thought to be a (Fe, Co, Ni)₃O₄ spinel. Then, followed by the fifth oxide layer is the region that is depleted in Fe and enriched in Ni and to a lesser extent Co.

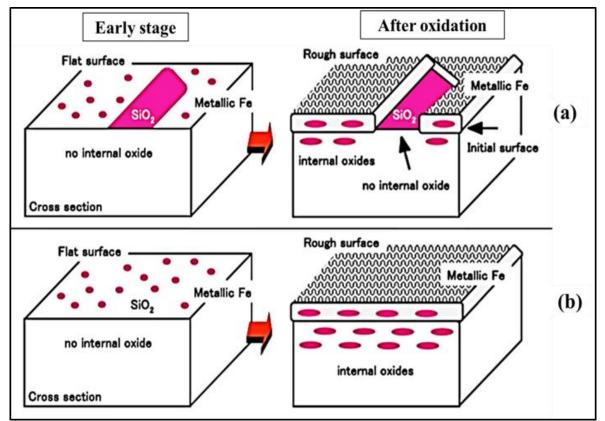


Fig. 5. Schematic surface structure of Fe-Si alloy samples, (a) A and (b) B caused by oxidation [34]



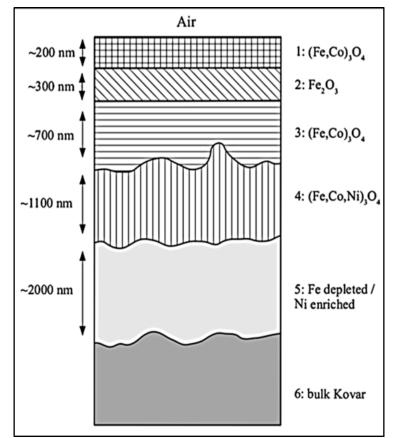


Fig. 6. The proposed structure of oxides developed on Kovar with approximate thickness grown at $700^{\circ}C$ [71]

Many studies have reported the phase transformation of FeO and as studied by Kondo and Tanei [41] the phase transformation of steel scale advances with Fe₃O₄ precipitating in the first place as shown in Figure 7. The Fe₃O₄ that precipitates at the beginning sometimes appears from interface between the base metal and FeO and this layer is called magnetite seam [97-98]. This transformation process oxidizing occurred at 700 to 750°C, where the scale comprised of FeO, Fe₃O₄ and Fe₂O₃. Meanwhile by holding at 550°C in N₂ atmosphere, the Fe₂O₃ layer at the surface was changed into Fe₃O4 and the scale comprised of two layers of FeO and Fe₃O₄. By holding at 700°C, since FeO exist stably the Fe₃O₄ and Fe₂O₃ layers changed into a single FeO layer and thus the layer only comprised of one FeO layer. The two-layer (bilayer) scale made up of FeO and Fe₃O₄ and single-layer (monolayer) scale of FeO only obtained before phase transformation [41,99-101].

It is important to note that, issues in steel welded joint that reported recurrently by previous researchers mostly regarding to grain growth, embrittlement, grain boundaries carbides and sigma phase etc. Due to such problems, mechanical properties like toughness, ductility, high temperature strength and corrosion resistance is seriously affected and it would lead to abnormalities in phase present [102]. These difficulties have seriously affected the use of this economical material in many structures. Many researchers have contributed enormously to improve the weldability of ferritic stainless steel. Therefore, it is suggested that more special attention is required for the improvement of mechanical properties of steel and alloys during welding which have still some issues as well as its prevention method applied.



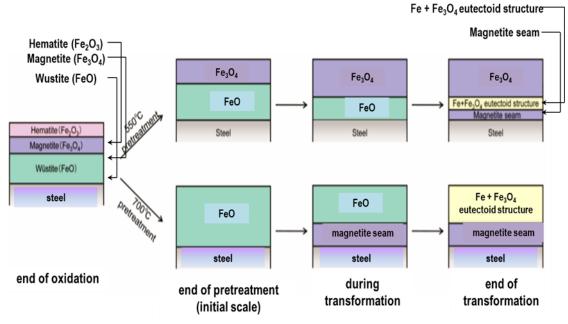


Fig. 7. The proposed structure of oxides developed on Kovar with approximate thickness grown at 700°C [71]

6. Conclusions

In this review, previous study on selective deposition and development of oxide layer has been focused on Fe-based alloys. The main compounds formed during reaction between Fe and oxygen when in contact with plenty of oxygen around is FeO. It is also known as wustite. Then, at high enough temperatures, the iron (III) oxide, Fe₂O₃ formed is hematite. Hematite presents in grey or reddish compound. When Fe is heated in the presence of less oxygen, it is possible to synthesize to form another iron oxide, Fe₃O₄ called magnetite and it occurs in black colour. Therefore, selective deposition that leads to an oxide phase transformation plays an important role in determining the oxide product characteristics. Future work needs to be done to measure and analyze the characteristics of each oxide products developed due to its heterogenous behavior in different oxidation environment and condition particularly at elevated temperatures.

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